

Surface-plasmon mode hybridization in subwavelength microdisk lasers

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Hybridization of surface-plasmon and dielectric waveguide whispering-gallery modes are demonstrated in a semiconductor microdisk laser cavity of subwavelength proportions. A metal layer is deposited on top of the semiconductor microdisk, the radius of which is systematically varied to enable mode hybridization between surface-plasmon and dielectric modes. The anticrossing behavior of the two cavity mode types is experimentally observed via photoluminescence spectroscopy and optically pumped lasing action at a wavelength of $\lambda \sim 1.3 \mu\text{m}$ is achieved at room temperature. © 2009 American Institute of Physics.

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In wavelength-scale lasers, the very small number of optical modes and small volume of gain material allows one to probe the subtle and often interesting properties of lasing action.¹ Semiconductor microdisk lasers, in particular, have been actively studied due to their simple geometry and amenability to planar chip-scale integration with microelectronics.²⁻⁴ More recently there has been great interest in using surface-plasmon (SP) modes at a semiconductor-metal interface for guiding as well as high intensity and sub-wavelength optical confinement.⁵ There has been significant work on the incorporation of SP waveguides that also act as electrical contacts in mid-infrared quantum cascade lasers,⁶ in increasing SP propagation lengths using SP-dielectric waveguide mode hybridization,⁷ as well as in creating ultrasmall laser cavities.^{8,9}

In miniaturizing semiconductor lasers to the nanoscale one encounters several design challenges that must be addressed, such as thermal management,^{10,11} proximity of metal contacts to the optical cavity, surface states,¹² and demanding tolerance levels in fabrication. In this letter, we investigate the purposeful integration of a metal contact into a subwavelength whispering-gallery microdisk laser. We show that whispering-gallery SP and dielectric modes hybridize into low loss modes. We predict and map out this hybridization using finite-element-method (FEM) simulations, and experimentally measure the properties of fabricated microdisk laser cavities with varying levels of mode hybridization.

Simulation of the hybrid laser cavities is performed using fully-three-dimensional FEM simulations with azimuthal symmetry.¹³ A 250 nm thick semiconductor disk with index $n_{\text{disk}}=3.4$ and radius $R_d=0.65 \mu\text{m}$ is simulated with a centered metal contact of varying radius (R_m). A schematic of the microdisk device is shown in Fig. 1(a). Silver with a complex refractive index of $n_{\text{Ag}}=0.11-i9.5$ at $\lambda=1.3 \mu\text{m}$ is chosen for the metal layer due to its low optical loss.¹⁴ For this disk size in the 1300 nm wavelength band there occurs a near degeneracy of the transverse-electric-like (TE-like) whispering-gallery mode (WGM) with dominant electric field polarization in the plane of the disk and the transverse-magnetic-like (TM-like) mode with dominant electric field

normal to the disk plane. A plot of the wavelength and Q -factor of these two resonances is shown in Fig. 2(a) as a function of metal radius fraction (R_m/R_d). The resonances exhibit a clear anticrossing behavior, with the modes hybridizing and picking up significant SP character with increased metal coverage. At one extreme, where R_m/R_d is very small, the upper wavelength branch (mode I) is of TE-like WGM character and the lower wavelength branch (mode II) is of TM-like WGM character. At the other extreme, $R_m/R_d \sim 1$, the upper branch has taken on a SP mode character whereas the lower branch is now TE-like. Dominant electric field components for both cases are plotted in Fig. 2(b).

Of particular interest would be the lower wavelength branch (mode II) of the SP-dielectric hybrid modes, as this mode shows significant robustness in its optical Q -factor for large metal coverage. Further analysis of this regime is performed by studying the effects of an InP pedestal ($n=3.2$)

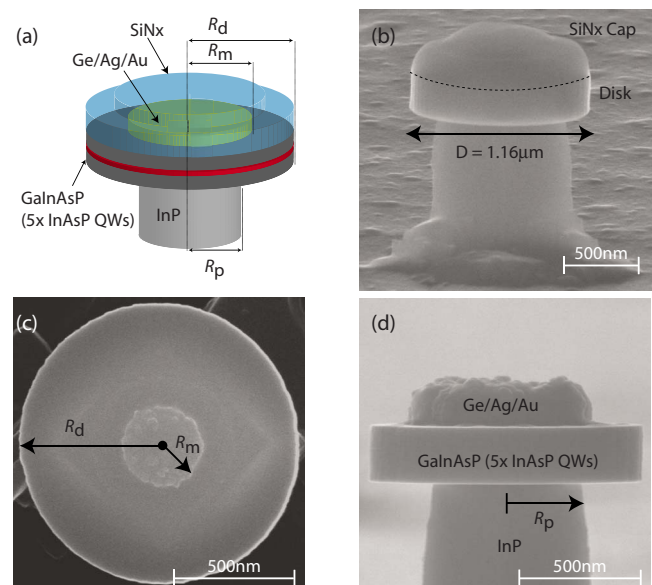


FIG. 1. (Color online) (a) Schematic of fabricated and simulated microdisks with a metal contact buried below a SiN_x cap. [(b)–(d)] SEM images of a diameter $D=1.16 \mu\text{m}$ fabricated microdisk with a Ge/Ag/Au 10/80/10 nm contact buried under a 150 nm conformal layer of SiN_x . Dashed line delineates GaInAsP– SiN_x boundary. (c) Top view and (d) cross-sectional view after SiN_x has been removed.

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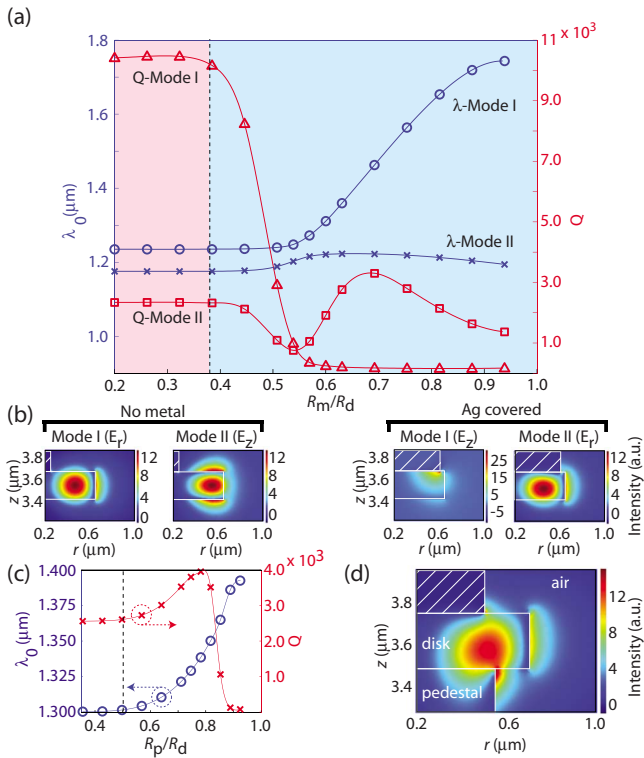


FIG. 2. (Color online) A 250 nm thick semiconductor microdisk with radius $R_d=0.65 \mu\text{m}$ and a top silver contact is simulated with varying silver coverage. (a) Wavelength (λ_0) and quality factor (Q) for two anticrossing modes (mode I and mode II) are plotted as a function of silver metal radial fraction (R_m/R_d). (b) Azimuthal slices of dominant electric field components of the two extreme cases: with and without silver. Metal is denoted by white hatch marks. (c) Wavelength and Q of mode II as a function of increasing InP pedestal radius (R_p) with $R_m/R_d=0.7$. (d) Mode profile of E_r field component of mode II with maximum Q ($R_p/R_d=0.8$).

used to support the microdisk. Figure 2(c) shows a plot of cavity Q of mode II versus the pedestal radius (R_p) with $R_d=0.7 \mu\text{m}$ so the resonant wavelength is $\lambda \approx 1.3 \mu\text{m}$. In this plot the metal-to-disk ratio was set at the optimal value of $R_m/R_d=0.7$ from Fig. 2(a). A peak value of $Q=4000$ is found for a fractional pedestal radius of $R_p/R_d=0.8$ [corresponding mode field plot shown in Fig. 2(d)]. From these simulations it is clear that a microdisk cavity structure with significant metal coverage and very little undercut can be designed to have a cavity Q -factor sufficient for lasing action. Such a structure would facilitate fabrication as well as good thermal heat sinking.

In order to test the predictions of the FEM-modeling, microdisks with nominal diameter $D=1.2 \mu\text{m}$ were fabricated from 252 nm thick membranes consisting of five InAsP/GaInAsP compressively strained quantum wells, with peak spontaneous emission at $\lambda \approx 1285 \text{ nm}$.¹⁵ First, the metal layer (Ge/Ag/Au=10/80/10 nm) was deposited and patterned into round contacts of systematically varying diameter by electron beam lithography (EBL), electron beam evaporation, and liftoff. A hard SiN_x mask was then deposited to protect the patterned metal layer, and the outer disk shape was patterned by a second, aligned EBL step. An inductively coupled reactive-ion etch was used to transfer the disk pattern through the hard mask and the 252 nm thick semiconductor layer. The disks were then undercut, and the pedestal formed, using HCl:H₂O solution.¹⁶ A scanning electron microscope (SEM) micrograph of a final device in-

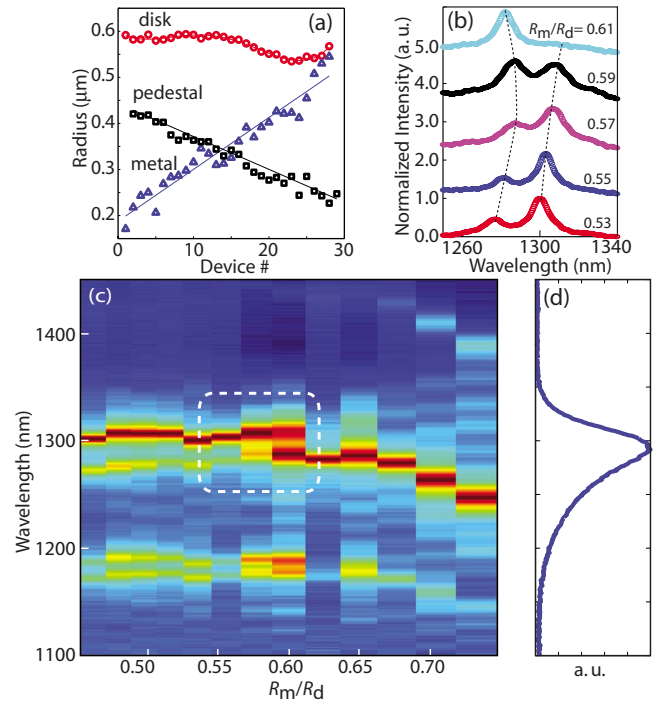


FIG. 3. (Color online) (a) Measured disk, metal, and pedestal radii of fabricated device array, (b) Zoom-in of anticrossing region as indicated by a dashed white box in panel (c). Dotted lines are guides to the eye. (c) Fiber-taper-collected normalized spectra on a log-scale as a function of R_m/R_d ($P_p=1 \text{ mW}$, $\delta T=500 \text{ ns}$, and $T=1 \mu\text{s}$). (d) Free-space-collected PL of unpatterned laser material ($P_p=282 \mu\text{W}$, cw).

cluding remaining SiN_x cap is shown in Fig. 1(b). After the device testing described below, the SiN_x cap layer was removed allowing for SEM imaging [Figs. 1(c) and 1(d)] and measurement of the disk, metal, and pedestal radii [Fig. 3(a)]. A systematic variation of 25%–100% metal coverage was achieved in disks with average diameter $D=1.2 \mu\text{m}$.

Initial resonance mode spectroscopy was performed by free space optical pumping and collection of the photoluminescence (PL) through an optical fiber taper nanoprobe.¹⁷ The fiber taper provides excellent collection efficiency of the WGM emission of the microdisk, substantially improving the sensitivity of the measurement. Using a pulsed external cavity diode laser at $\lambda=830 \text{ nm}$, the disks were pumped with pulses of peak power $P_p \approx 1 \text{ mW}$, pulse width $\delta T=500 \text{ ns}$, and pulse period $T=1 \mu\text{s}$. The fiber-collected spectra are plotted in Fig. 3(c) for devices with varying R_m/R_d . Two pairs of modes can be seen at $\lambda \sim 1.2$ and $1.3 \mu\text{m}$. A normalized PL spectrum taken with continuous wave pumping from the unprocessed semiconductor material ($P_p=282 \mu\text{W}$) is shown in Fig. 3(d) for reference. Despite the disappearance of the upper wavelength branch mode in the spectra of Fig. 2(c) for large metal coverage, a result expected due to the very low- Q of the SP mode, the zoom-in shown in Fig. 3(b) for the longer-wavelength mode pair shows clear anticrossing behavior indicative of mode hybridization of the SP and dielectric WGMs. A quantitative comparison of the modal Q for the different resonant modes is hindered by the wavelength and pump dependent nature of the quantum-well active region of these laser devices.

To study laser action in these hybridized cavities the fiber taper is removed to eliminate external cavity loading effects, and vertically scattered light emission from the microdisks is collected via a high numerical aperture lens in-

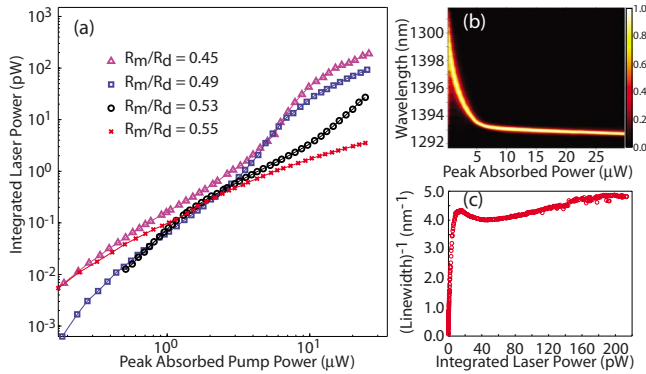


FIG. 4. (Color online) (a) L-L curves of several devices of increasing R_m/R_d . (b) Normalized spectra vs pump power, and (c) inverse linewidth vs output laser power for a device with $R_m/R_d=0.42$.

stead. The microdisks are also pumped with low-duty-cycle pulses ($\delta T=20$ ns, $T=4$ μ s) in order to reduce thermal effects. Threshold curves for a series of microdisks near the region of strong SP and dielectric mode hybridization are shown in Fig. 4(a). The peak absorbed pump power is estimated based on a pump spot size of diameter $D=2$ μ m, absorption efficiency of $\eta=10\%$, and accounting for the pump duty cycle. Disks with $R_m/R_d < 0.53$ exhibit a clear “S” shaped logarithmic light-in versus light-out curve, indicating lasing action. In each of these cases, the lasing mode is from the longer wavelength branch of the hybridized modes. Typical of the laser behavior for these devices is the laser with $R_m/R_d=0.42$, which has an estimated threshold peak absorbed pump power of only $P=5$ μ W. Spectra as a function of peak absorbed pump power for this laser are plotted in Fig. 4(b). A strong blueshift of the laser wavelength with increased pumping is seen, attributable to free-carrier dispersion. The inverse linewidth as a function of integrated output power is also plotted in Fig. 4(c), and shows linewidth narrowing typical of a semiconductor laser with large coupling between carrier density (gain) and refractive index (cavity frequency).^{18,19} In the middle of the anti-crossing region, for microdisks with $0.55 \leq R_m/R_d \leq 0.65$, no lasing was observed. This is likely due to the increased optical loss in this region predicted by simulation [see Fig. 2(a)]. For microdisks with $R_m/R_d > 0.65$ lasing action was also not observed, even for the higher- Q shorter wavelength branch of modes. Although simulations indicate that the optical cavity Q -factor for this mode should recover (and increase for an optimal pedestal size), as can be seen in Figs. 3(c) and 3(d) the mode blueshifts a significant fraction of the PL bandwidth. As the devices that do lase are pumped quite hard to reach the lasing condition, any reduction in gain or Q could preclude lasing action.

Beyond the initial demonstration of SP-dielectric mode hybridization and lasing in subwavelength partially metal coated microdisk cavities presented here, future efforts will focus on achieving electrically injected lasing action using

the surface-plasmon metal layer as a top metal contact and thermal heat sink. It is anticipated that further device engineering involving the top metal contact layer should allow room temperature, continuous-wave (cw) lasing action in such subwavelength laser cavities. An improved optical quality factor may also be achieved in the shorter-wavelength mode branch by engineering the epitaxy thickness. Ultimately, such improved structures should allow for the study of fundamental as well as practical issues associated with the scaling laws of deep-subwavelength mode volume semiconductor lasers, where the small-scale system size of the laser results in significant photon number and carrier number fluctuations.²⁰

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- ¹Y. Yamamoto and R. E. Slusher, *Phys. Today* **46**(6), 66 (1993).
- ²A. F. J. Levi, S. L. McCall, S. J. Pearton, and R. A. Logan, *Electron. Lett.* **29**, 1666 (1993).
- ³T. Baba, M. Fujita, A. Sakai, M. Kihara, and R. Watanabe, *IEEE Photonics Technol. Lett.* **9**, 878 (1997).
- ⁴O. Painter, R. K. Lee, A. Yariv, A. Scherer, J. D. O’Brien, P. D. Dapkus, and I. Kim, *Science* **284**, 1819 (1999).
- ⁵W. L. Barnes, A. Dereux, and T. W. Ebbesen, *Nature (London)* **424**, 824 (2003).
- ⁶C. Sirtori, C. Gmachl, F. Capasso, J. Faist, D. L. Sivco, A. L. Hutchinson, and A. Y. Cho, *Opt. Lett.* **23**, 1366 (1998).
- ⁷B. K. Min, E. Ostby, V. Sorger, E. Ulin-Avita, L. Yang, X. Zhang, and K. Vahala, *Nature (London)* **457**, 455 (2009).
- ⁸M. T. Hill, Y. S. Oei, B. Smalbrugge, Y. Zhu, T. D. Vries, P. J. Veldhoven, F. W. M. V. Otten, J. P. Turkiewicz, H. D. Waardt, E. J. Geluk, T. J. Eijkemans, S.-H. Kwon, Y.-H. Lee, R. Nötzel, and M. K. Smit, *Nat. Photonics* **1**, 589 (2007).
- ⁹M. T. Hill, M. Marell, E. S. P. Leong, B. Smalbrugge, Y. C. Zhu, M. H. Sun, P. J. van Veldhoven, E. J. Geluk, F. Karouta, Y. S. Oei, R. Nötzel, C.-Z. Ning, and M. K. Smit, *Opt. Express* **17**, 11107 (2009).
- ¹⁰M. H. Shih, M. Bagheri, A. Mock, S. J. Choi, J. D. O’Brien, P. D. Dapkus, and W. Kuang, *Appl. Phys. Lett.* **90**, 121116 (2007).
- ¹¹J. K. Hwang, H. Y. Ryu, D. S. Song, I. Y. Han, H. K. Park, D. H. Jang, and Y. H. Lee, *IEEE Photonics Technol. Lett.* **12**, 1295 (2000).
- ¹²K. Tai, T. R. Hayes, S. L. McCall, and W. T. Tsang, *Appl. Phys. Lett.* **53**, 302 (1988).
- ¹³M. Borselli, T. J. Johnson, and O. Painter, *Appl. Phys. Lett.* **88**, 131114 (2006).
- ¹⁴P. B. Johnson and R. W. Christy, *Phys. Rev. B* **6**, 4370 (1972).
- ¹⁵W.-Y. Hwang, J. Baillargeon, S. N. G. Chu, P. F. Sciortino, and A. Y. Cho, *J. Vac. Sci. Technol. B* **16**, 1422 (1998).
- ¹⁶K. Srinivasan, P. E. Barclay, O. Painter, J. Chen, and A. Y. Cho, *J. Vac. Sci. Technol. B* **22**, 875 (2004).
- ¹⁷C. P. Michael, M. Borselli, T. J. Johnson, C. Chrystal, and O. Painter, *Opt. Express* **15**, 4745 (2007).
- ¹⁸G. P. Agrawal and G. R. Gray, *Appl. Phys. Lett.* **59**, 399 (1991).
- ¹⁹G. Björk, A. Karlsson, and Y. Yamamoto, *Appl. Phys. Lett.* **60**, 304 (1992).
- ²⁰P. R. Rice and H. J. Carmichael, *Phys. Rev. A* **50**, 4318 (1994).